

## PARAMETRIC STUDY OF REACTIVE MELT INFILTRATION

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### ABSTRACT

Reactive melt infiltration is viewed as a promising means of achieving near-net shape manufacturing with quick processing time and at low cost. Since the reactants and products are, in general, of varying density, overall conservation of mass dictates that there is a force related to chemical conversion which can directly influence infiltration behavior. In effect, the driving pressure forces may compete with the forces from chemical conversion, affecting the advancement of the front. We have developed a two-dimensional numerical code to examine these effects, using reaction-formed silicon carbide as a model system for this process. We have examined a range of initial porosities, pore radii, and reaction rates in order to investigate their effects on infiltration dynamics.

### NOMENCLATURE

#### Roman letters

$k_r$	reaction rate constant
$MW_i$	species molecular weight
$p$	pressure
$r$	pore radius
$R_i$	rate of production/consumption of species $i$
$t_{e,i}$	time at which infiltration is complete
$t_{e,r}$	time at which chemical conversion is complete
$t_3$	time at which infiltration enters third, post-reaction-dominated regime
$\tilde{u}_D$	macroscopic Darcy (front) velocity
$\bar{v}_i$	species molar specific volume
$V$	volume
$y_f$	location of the infiltration front

#### Greek letters

$\xi$	intrinsic infiltration rate
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$\kappa$	permeability
$\eta$	molar consumption rate
$\phi_i$	species volume fraction, $V_i/V$
$\Phi$	porosity
$\Phi_s$	volume fraction of solid, $\Phi_s = 1 - \Phi$
$\sigma$	surface tension
$\theta$	wetting angle
$\mu$	fluid viscosity

#### Subscripts/superscripts

$c$	capillary
$f$	front
$i$	index denoting species

### BACKGROUND

In reactive melt infiltration (RMI), a porous medium, or "preform", is first created, which is easy to form into a desired shape. A liquid infiltrant is introduced into the preform through a pressure gradient caused by capillary action or by applied pressure (see, e.g., Kochendoerfer, 1991; Hillig, 1994a, b; Haggerty and Chiang, 1990). The chemical conversion between the infiltrant and the preform results in a final product, which may be fully dense and is of the same dimensions as the original preform (Singh and Behrendt, 1992).

In comparison to other methods of composite manufacture, RMI is quick and efficient. Chemical vapor infiltration requires days or weeks to complete the chemical conversion process, but RMI needs only minutes or hours. Post-process machining is limited, making this process attractive for the manufacture of brittle and expensive ceramics.

Reaction-formed silicon carbide is chosen as a model system for this study. The production of monolithic silicon carbide system is appealing because the chemistry and preform geometry are reasonably well characterized. We will focus on infiltration that is driven solely by capillary pressure. The relevant thermophysical properties of the chemical constituents are given in Table 1.

**Table 1. Thermophysical properties of silicon, carbon, and silicon carbide.**

Property	Si	C	SiC
$\rho$ (kg/m <sup>3</sup> )	2400	1500	3120
$\mu$ (kg/m-s)	$5 \times 10^{-4}$		
MW (kg/kmol)	28.1	12.0	40.1
$\bar{v}_i$ (m <sup>3</sup> /kmol)	0.0117	0.008	0.0128

Although the absolute viscosity of liquid silicon,  $\mu$ , is a weak function of temperature (Turovskii and Ivanova, 1974), we will neglect any temperature dependence. Note that the specific molar volume,  $\bar{v}_i$ , varies substantially among the species. In fact, the volume occupied by a mole of SiC is approximately 35% smaller than that of a mole each of Si and C. Based on overall mass conversion, Singh and Behrendt computed that the critical porosity,  $\Phi_c$ , for this process is 0.358. For initial porosities below  $\Phi_c$ , the flow will choke prematurely because there simply is not enough volume available to complete the conversion. If  $\Phi_0 > \Phi_c$  and all of the carbon is converted in a fixed preform volume, the resulting ceramic will be comprised of SiC and Si. In this study, we will investigate  $\Phi_0$  in the range of 0.40-0.60, well above the critical porosity.

In order to model the preform, we have drawn on the extensive work of Behrendt and Singh on the melt infiltration of silicon and silicon alloys into microporous carbon (see Behrendt, 1986; Behrendt and Singh, 1993, 1994; Singh and Behrendt, 1992, 1994a, b, 1995). Their work was also the basis for other experimental and analytical studies (Sangsuwan et al., 1998; Datta et al., 1996; Messner and Chiang, 1990). The amorphous, glassy carbon preforms were uniform and isotropic with good connectivity (Singh and Behrendt, 1992, 1994b; Sangsuwan et al., 1998). Moreover, they could be consistently reproduced by Behrendt and Singh's process with pore diameters ranging from 0.1-3.0  $\mu\text{m}$  and initial porosities from 0.33-0.64. We will examine pore radii ranging from 0.10-0.20  $\mu\text{m}$  and initial porosities from 0.40-0.60.

Infiltration takes place in an evacuated furnace. Without the application of external pressure, the infiltration process is driven by capillary action. Consequently, the infiltration rate is dependent upon the wetting characteristics of the infiltrant

upon the preform surface, which is in turn sensitive to local temperature and surface and melt impurities. At 1703 K, the contact angle,  $\theta$ , of liquid silicon on carbon is reported to be 38° by Li and Hausner (1991). (See also Whalen and Anderson, 1974; Turovskii and Ivanova, 1974.) We will use a value of 0.725 N/m for the surface tension,  $\sigma$  (Brandes, 1983), although other values have been reported (Minnear, 1982).

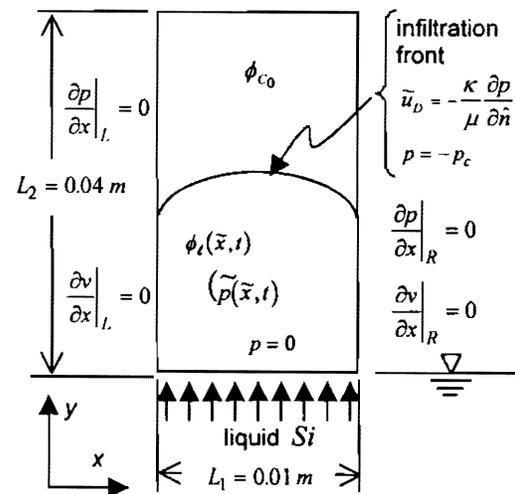
We will assume that the chemistry is reaction-controlled rather than diffusion-dominated, based on the work of Messner and Chiang (1990) and Fitzer and Gadov (1984) so that the pore radii close by:

$$r(t) = r_0 - k_r t_r \quad (1)$$

where  $r_0$  is the initial pore radius,  $k_r$  is the reaction rate, and  $t_r$  is the local reaction time, i.e., the amount of time that has elapsed since the reaction began in this region. Since the temperature variation within the preform for this set of parameters is on the order of 100°C (Nelson, 1998; Einset, 1993), we will use a constant value for  $k_r$ . Messner and Chiang estimated the reaction rate to be  $4 \times 10^{-8}$  m/s, but bounded the estimate from  $10^{-8}$  to  $10^{-7}$  m/s. Alloyed melts may decrease reaction rates as well as eliminate the residual silicon phase (Singh and Behrendt, 1992, 1994a, b, 1995). Accordingly, we will investigate reaction rates in the range of  $10^{-9}$  to  $10^{-7}$  m/s. In reality, the reaction rate is not a tunable parameter, and the effect on infiltration dynamics will vary if the chemistry is modified.

## GOVERNING EQUATIONS

A schematic of the two-dimensional numerical model of reactive melt infiltration is sketched below in Fig. 1. Liquid Si



**Figure 1. Schematic of reactive melt infiltration.**

is introduced at the bottom of the preform and wicks up into the preform due to capillary action.

The general form for the conservation of mass is given for each of the species by:

$$\frac{\partial}{\partial t}(\rho_i \phi_i) + \nabla \cdot (\rho_i \phi_i \tilde{u}_i) = R_i \quad i = Si, C, SiC \quad (2)$$

where  $R_i$  is the rate of production/destruction of species  $i$ , and  $\phi_i$  is the species volume fraction, i.e., the fraction of the volume occupied by species  $i$ . The species volume fraction must obey the constraints  $0 \leq \phi_i \leq 1$  and  $\sum_i \phi_i \leq 1$ . In any region that is completely filled,  $\sum_i \phi_i = 1$ . Since overall mass must be conserved, the conservation equation for the total mass in the system is subject to the constraint  $\sum_i R_i = 0$ . The liquid phase, comprised of a single component, i.e.,  $Si$ , is the only phase with a nonzero velocity.

We will model this process as a zeroth-order chemical reaction. For stoichiometric conversion, this exothermic reaction is given by  $Si + C \rightarrow SiC$ , so that the local production of species can be expressed as:

$$R_i = \beta \eta MW_i \quad (3)$$

where  $\beta = +1$  for the product and  $-1$  for the reactants,  $\eta$  is the number of moles consumed per unit time, and  $MW_i$  is the species molecular weight. We note that none of the species can be updated to a negative value. Since we are provided with an abundant source of  $Si$ , the amount of  $C$  available locally will limit the chemical conversion. Therefore, when all of the  $C$  has been consumed locally, the chemical reaction comes to a halt in that region.

We will make the critical assumption that the overall volume of the preform remains constant (Singh and Behrendt, 1992). Using the above assumptions, the overall conservation of mass can be written as:

$$\nabla \cdot (\phi_{Si} \tilde{u}_{Si}) = \eta (\bar{u}_{SiC} - \bar{u}_{Si} - \bar{u}_C) \quad (4)$$

where  $\bar{u}_i = MW_i / \rho_i$  is the specific molar volume.

The expression for  $\eta$  is found by relating the geometric description of the pore closure to that obtained from the conservation of mass. The volume fraction of the solid phase is  $\Phi_s = \phi_C + \phi_{SiC}$ . From mass conservation, the change in solid volume fraction with respect to time is:

$$\begin{aligned} \frac{\partial \Phi_s}{\partial t} &= \frac{\partial \phi_C}{\partial t} + \frac{\partial \phi_{SiC}}{\partial t} \\ &= \eta (\bar{u}_{SiC} - \bar{u}_C) \end{aligned} \quad (5)$$

If we envision the preform porosity to be comprised of a bundle of parallel capillary tubes, then the permeability for this model is found by:

$$\kappa = \frac{\Phi r^2}{8} \quad (6)$$

where  $r$  is the pore radius and  $\Phi$  is the porosity ( $\Phi_s = 1 - \Phi$ ). Experimental data (Datta et al., 1996) indicates that this is an adequate model for this preform. Comparing the cross-

sectional area of pore space initially and at time  $n$ , we find that:

$$\frac{\partial \Phi_s}{\partial t} = \frac{\partial}{\partial t} \left( \frac{\Phi_0}{r_0^2} (r_0^2 - r(t_r)^2) \right) \quad (7)$$

For details, see Nelson, 1998; Nelson and Colella, 1999a, b. Recall that  $r(t) = r_0 - k_r t_r$ , where  $r_0$  is the initial pore radius and  $t_r$  is the relevant reaction time. Combining Eqs. (5) and (7) gives the molar consumption rate as:

$$\eta = \frac{2}{\bar{u}_{SiC} - \bar{u}_C} \frac{k_r}{r_0} \Phi_0 \left( 1 - \frac{k_r}{r_0} t_r \right) \quad (8)$$

At the pore level, the convective time scale is much longer than the diffusive time scale. Therefore, we can use d'Arcy's law to describe the motion of the front:

$$\tilde{u}_D = -\frac{\kappa}{\mu} \nabla p \quad (9)$$

where  $\tilde{u}_D$  is the front velocity,  $\kappa$  is the permeability of the porous medium,  $\mu$  is the absolute viscosity of the fluid, and  $p$  is the pressure. For this system,  $\kappa$  is a function of space and time. Furthermore, any body force term is negligible in comparison to the pressure force provided by capillary action.

We can relate the fluid velocity,  $\tilde{u}_{Si}$ , to the front velocity,  $\tilde{u}_D$ , by the Dupuit-Forcheimer relation, i.e.,  $\tilde{u}_D = \phi_{Si} \tilde{u}_{Si}$ . We do not explicitly know  $\tilde{u}_D$  or the pressure field; however, our analysis of continuity provided a relationship for the divergence of velocity in Eq. (4). Combining the divergence of d'Arcy's law, Eq. (9), with Eq. (4), we arrive at a steady Poisson problem for pressure that stems from the conservation of momentum and of mass:

$$\nabla \cdot \left( -\frac{\kappa}{\mu} \nabla p \right) = \eta (\bar{u}_{SiC} - \bar{u}_{Si} - \bar{u}_C) \quad (10)$$

Equation (10) identifies the mechanism by which capillary forces and chemical conversion can enter into competition in the infiltration process. For this system, the right-hand side of Eq. (10) is at minimum zero. When it is nonzero, the pressure profiles within the preform exhibit some curvature, resulting in a nonuniform velocity field.

Since the vapor pressure of liquid silicon is very small and the experimental furnace is evacuated, we assume that the ambient pressure is negligible. We apply a homogeneous Dirichlet boundary condition at the lower boundary; at the infiltration front, the pressure is equal to the negative of the capillary pressure,  $-p_c$ . Using the Young-Laplace equation, we can estimate  $p_c$  at the front as a function of surface tension,  $\sigma$ , pore radius,  $r$ , and the contact angle,  $\theta$ :

$$p_c = \frac{2\sigma \cos \theta}{r} \quad (11)$$

Depending on the initial pore radius,  $p_c$  varied from about  $5.94 \times 10^6$  to  $1.88 \times 10^7$  Pa. As in the experimental setup of Sangsuwan et al. (1998), we will assume that the preform is coated with, e.g., boron nitride, to inhibit mass transport across

the sidewalls. This ensures that the liquid silicon from the bath will not wick up the sidewalls, and that no fluid will escape from the system through the sidewalls. Therefore, homogeneous Neumann boundary conditions are applied to the pressure at the side boundaries.

After the pressure solve, we can directly calculate the velocity from the pressure field by using d'Arcy's law, Eq. (9). The normal component of velocity at the sidewalls is zero, due to the homogenous Neumann boundary condition used on the pressure field. Although we would expect some tangential velocity slip in the immediate vicinity of the sidewalls, these effects are confined to distances on the order of a pore diameter (Kaviani, 1991). Therefore, we will assume that, at the left and right boundaries,  $\partial v/\partial x = 0$ . In this set of numerical simulations, the infiltration process is essentially one-dimensional, as in experimental work of a similar process (Einset, 1993). This can be expected due to uniform initial porosity and pore size, the choice of boundary conditions on pressure and velocity, and the simplicity of the chemistry and permeability models.

The derivation of the energy equation, as well as the full details of the above derivations, can be found in Nelson (1998) or Nelson and Colella (1999a, b).

## NUMERICAL PROCEDURE

The equations derived above are used to develop a finite-volume algorithm for reactive melt infiltration. The discrete formulation in two dimensions is discussed in Nelson (1998) and Nelson and Colella (1999a). Briefly, the macroscopic velocity is used to advance the front on a fixed mesh using a front-tracking technique developed by Chern and Colella (1988), coupled with a volume-of-fluid method and front reconstruction. Chemical reaction is allowed to occur based on the local species volume fractions and local reaction time. The updated species volume fractions are used to form a new permeability field. Next, a Poisson problem is solved on the infiltrated region only to find the new pressure field, using the technique of Johansen and Colella (1998). Finally, d'Arcy's law is used to find the new velocity field. The process is repeated until all of the C in the preform is consumed or the final simulation time is reached.

The mesh size was 32x128 on a 0.01x0.04 m domain so that the mesh spacing  $\Delta x = \Delta y = 0.0003125$  m. A description of the discrete equations, as well as the effects of grid resolution and other sources of numerical error are discussed in Nelson and Colella (1999a) and Nelson (1998).

## RESULTS

In this section, we will explore the effects of varying reaction rate, initial porosity and initial pore radius.

### Effect of reaction rate

With the initial porosity fixed at 0.50 and the initial pore radius at 0.17  $\mu\text{m}$ , the infiltration curves for reaction rates of 0,  $10^{-9}$ ,  $10^{-8}$ , and  $10^{-7}$  m/s are shown below in Fig. 2. The capillary pressure corresponding to this case is  $6.99 \times 10^6$  Pa. Fig. 2(a) presents the location of the front at the centerline,  $y_f(t)$ , while Fig. 2(b) shows  $y_f^2(t)$ .

In the absence of chemical reaction,  $y_f(t)$  exhibits a square-root dependence, and the slope of  $y_f^2(t)$ , called the intrinsic infiltration rate (Hilgig, 1994b) and denoted as  $\xi$ , is constant. The front reaches the upper boundary of the preform, located at 0.04 m, at  $t_f = 15.51$  s.

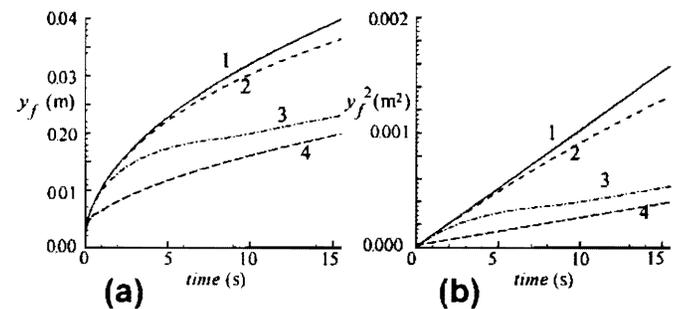


Figure 2. Centerline infiltration profiles for  $\Phi_0 = 0.50$ ;  $r_0 = 0.17 \mu\text{m}$ ;  $k_r = 0$  (1),  $10^{-9}$  (2),  $10^{-8}$  (3), and  $10^{-7}$  (4) m/s. (a)  $y_f(t)$ ; (b)  $y_f^2(t)$

For the chemical system of Si/C/SiC, the infiltration process is retarded by the chemical reaction. For the low reaction rate,  $k_r = 10^{-9}$  m/s, the infiltration curves closely match the nonreacting case for the first couple of seconds. After that point, the front begins to lag behind the nonreacting case. Fig. 2(b) shows that that  $\xi$  is not constant over this time interval. In contrast,  $y_f^2$  for  $k_r = 10^{-7}$  m/s appears quite linear after half a second. For  $k_r = 10^{-8}$  m/s, there is a short period of under one second during which the curves closely follow the nonreacting profiles. Up to about 8 seconds, the slope of  $y_f^2$  is continuously changing. After that time,  $\xi$  appears constant, nearly matching that of  $k_r = 10^{-7}$  m/s.

What explains this behavior? The infiltration moves through a sequence of flow regimes in which the forces induced by capillary action and by chemical reaction are in competition (Nelson, 1998; Nelson and Colella, 1999a, b). Initially, the pressure forces dominate as the infiltrant impulsively moves into the preform, indicated through the close correspondence of  $y_f$  with the nonreacting case. This can be seen in approximately the first two seconds of  $k_r = 10^{-9}$  m/s and the first second of  $k_r = 10^{-8}$  m/s.

In the second regime, the forcing term provided by the chemical reaction flares up and wanes. Within the preform,

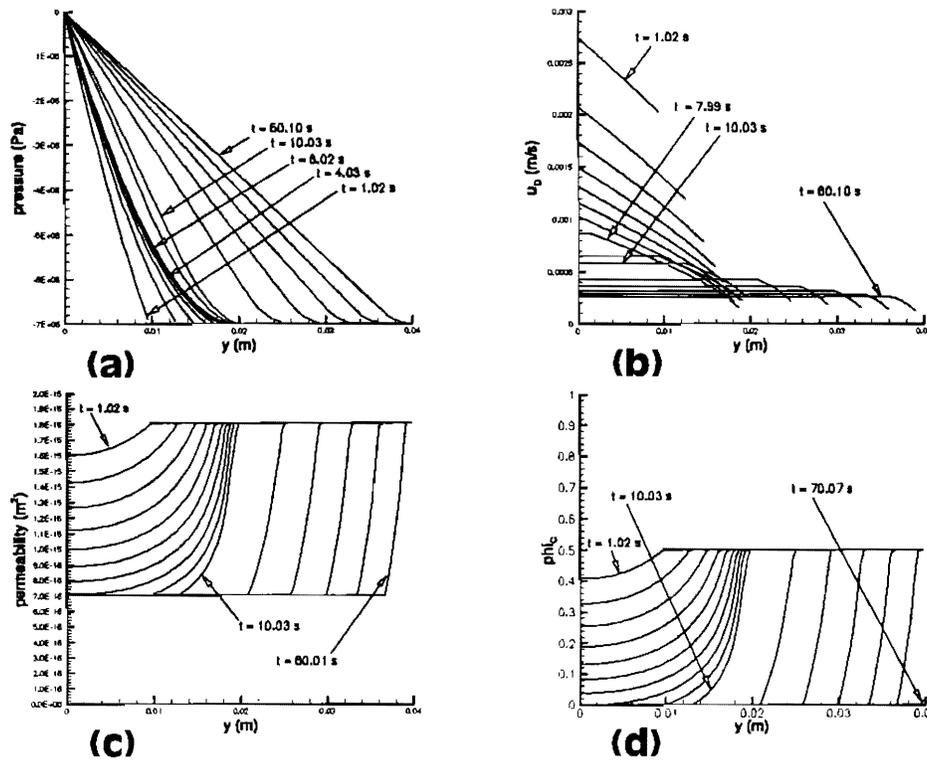


Figure 3. Centerline profiles of (a) pressure; (b) velocity; (c) permeability; and (d)  $\phi_c$  for  $\Phi_0 = 0.50$ ,  $r_0 = 0.17 \mu\text{m}$ , and  $k_r = 10^{-8}$  m/s. Profiles are presented in approximately one-second increments for  $t = 1$ -5 seconds, and in ten-second increments thereafter.

the centerline pressure profiles become progressively more curved (see Eq. (13)), leading to decreased flow velocities (see Eq. (9)), as shown in Figs. 3(a) and (b) for  $k_r = 10^{-8}$  m/s. Behind the front, the permeability,  $\kappa$ , Fig. 3(c), and  $\phi_c$ , Fig. 3(d), steadily decrease. Previous work (Nelson, 1998, Nelson and Colella, 1999a, b) indicates that the centerline thermal maximum lags well behind the infiltration front in this regime. Furthermore, the maximum operating temperatures were observed in this regime, particularly with insulated boundaries nearby.

When all of the available  $C$  has been consumed near the entry to the preform, Fig. 3(d) at  $t = 7.99$  s, the pressure relaxes to a more linear profile except in the immediate vicinity of the front, Fig. 3(a). Correspondingly, the velocity profile becomes constant over part of the infiltrated region, Fig. 3(b). Consequently,  $y_f(t)$  exhibits the characteristic bump in the infiltration profiles of Fig. 2, e.g., in  $k_r = 10^{-8}$  m/s between 4 and 8 seconds. This behavior can be observed in experimental work by Einset (1996).

For  $k_r = 10^{-9}$  m/s, the infiltration is still early in the reaction-dominated regime at 15 seconds, while for  $k_r = 10^{-7}$  m/s, the infiltration has passed through the reaction-dominated regime

by about one-half second. As will be seen throughout this work, the temporal and physical extent of the reaction-dominated regime decreases with increasing reaction rate.

After 10 seconds, the infiltration process for  $k_r = 10^{-8}$  m/s enters a third regime, in which the reaction-generated forces are confined to a small region behind the front and the infiltration dynamics are largely set by the permeability in the fully reacted zone and the capillary pressure. From 10 seconds to complete infiltration at 63 seconds,  $y_f(t)$  resembles a square-root profile, and  $\xi$  can be approximated by a constant. The reaction zone becomes progressively more localized to the vicinity of the front. For example, the reacting zone at 63 seconds is about 0.00296 m, appearing over eight computational cells. As a result, the thermal maximum is very close to the infiltration front (Nelson, 1998).

Although the curve for the highest reaction rate in Fig. 2(b) appears linear after one second, notice that, unlike the nonreacting case, the curve does not intersect the origin. Examination of the initial time interval more closely indicates that the infiltration simply proceeds through the reaction-dominated regime more rapidly than the other cases, resulting in the offset. If we were to extend the temporal window for

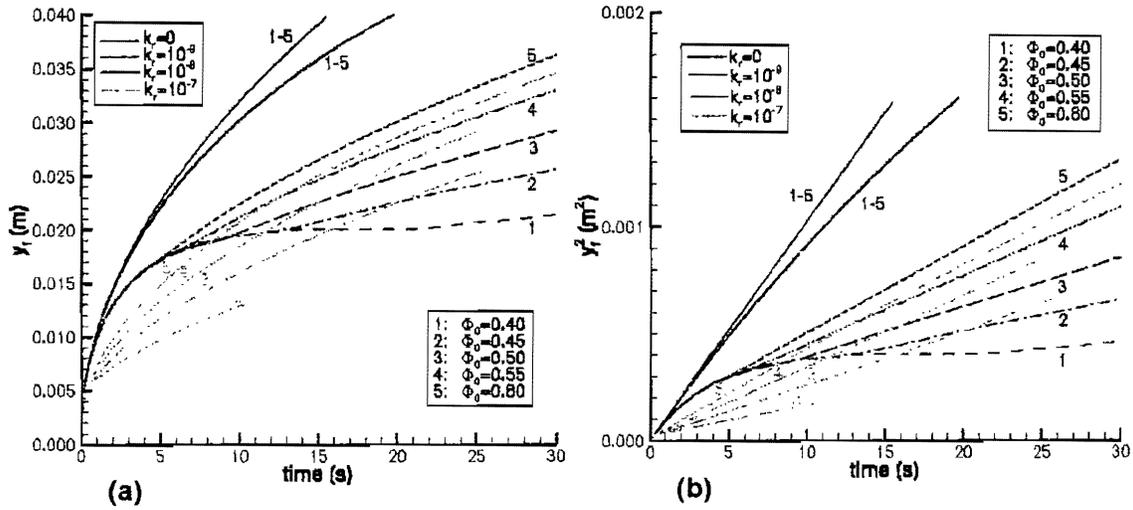


Figure 4. Effect of  $\Phi_0$  on infiltration profiles for  $r_0 = 0.17 \mu\text{m}$  and  $\Phi_0 = 0.40$  (1), 0.45 (2), 0.50 (3), 0.55 (4) and 0.60 (5):  $k_r = 0$  (black),  $10^{-9}$  (blue),  $10^{-8}$  (green), and  $10^{-7}$  m/s (red): (a)  $y_f$  vs.  $t$ ; (b)  $y_f^c$  vs.  $t$ .

$k_r = 10^{-9}$  m/s to later times (and with a longer preform length, in most cases), we would also see the characteristic bump signaling the ascendance and conclusion of the reaction-dominated regime, but occurring at a retarded pace.

When fully reacted, the final value for  $\phi_{Si}$  is 0.197 and for  $\phi_{SiC}$ , it is 0.803.

#### Effect of initial porosity

To examine the effect of varying porosity, infiltration profiles are shown in Fig. 4 for  $k_r = 0, 10^{-9}, 10^{-8}$ , and  $10^{-7}$  m/s with  $\Phi_0 = 0.40, 0.45, 0.50, 0.55$  and 0.60, holding  $r_0$  fixed at  $0.17 \mu\text{m}$ . The capillary pressure corresponding to this  $r_0$  is  $6.99 \times 10^6$  Pa. In this time interval, the infiltration profiles for  $k_r = 0$  and  $10^{-9}$  m/s collapse into two curves. The increased permeability caused by increasing the porosity by 50% does not significantly impact the infiltration in the pressure-dominated regime and the early stages of the reaction-dominated regime.

At increased reaction rate, we can still see an early period during which the curves for the varying porosity are indistinguishable. However, in the subsequent reaction-dominated regime, Fig. 4 indicates that the reaction-generated forces can greatly slow down the infiltration, as seen most dramatically for  $k_r = 10^{-8}$  m/s at the lowest porosities. In fact, the reaction-generated forces can theoretically stop the flow for a time. The fanning effect seen for  $k_r = 10^{-8}$  and  $10^{-7}$  m/s can be explained as follows: Since carbon is the only species present in the preform initially,  $\Phi_0$  is related to the amount of C present, i.e.,  $\phi_{C_0} = 1 - \Phi_0$ . This in effect places an upper bound on the amount of chemical conversion that is possible over a given volume. Once the carbon in the initial reaction-

dominated region is consumed, the dynamics of the infiltration proceed at a different pace. At constant  $k_r$ , examination of the centerline profiles reveal that the fanning occurs soon after the pressure profile relaxes back from its most distorted shape. These trends indicate that the reaction-dominated regime becomes larger in spatial and temporal extent with decreasing porosity and reaction rate. For the fully reacted zone, the final  $\phi_{SiC}$  was 0.96, 0.88, 0.80, 0.72, and 0.64 for  $\Phi_0 = 0.40, 0.45, 0.50, 0.55$  and 0.60, respectively.

The times required for complete infiltration and complete conversion,  $t_{e,i}$  and  $t_{e,r}$ , are presented in Fig. 5. We did not

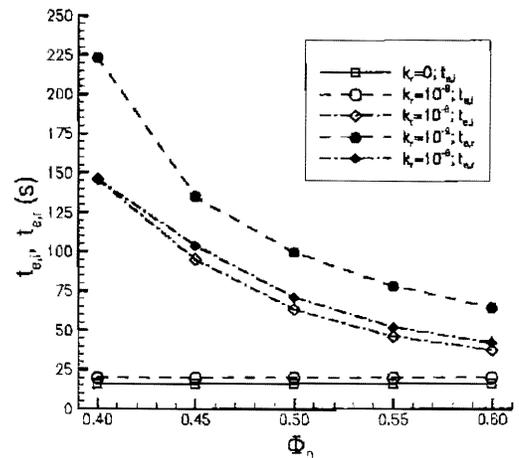


Figure 5. Influence of  $\Phi_0$  on required time for complete infiltration (unfilled symbols) and conversion (filled symbols) for  $r_0 = 0.17 \mu\text{m}$  and  $k_r = 0, \square; 10^{-9}, \bullet; 10^{-8}$  m/s,  $\blacklozenge$ .

carry out the infiltration process to complete conversion for  $k_r=10^{-7}$  m/s due to numerical instability; this is being addressed in a modified algorithm (Nelson and Colella, 1999a).

Notice that the most efficient reaction rate for infiltrating the preform, i.e., the lowest  $k_r$ , does not imply the shortest conversion time, particularly at low initial porosities. Also note that, as the reaction rate is increased at fixed porosity, the difference between  $t_{e,c}$  and  $t_{e,l}$  decreases. It is zero for  $\Phi_0=0.40$ ,  $k_r=10^{-8}$  m/s, indicating that the carbon behind the front is essentially consumed as fast as the front can move.

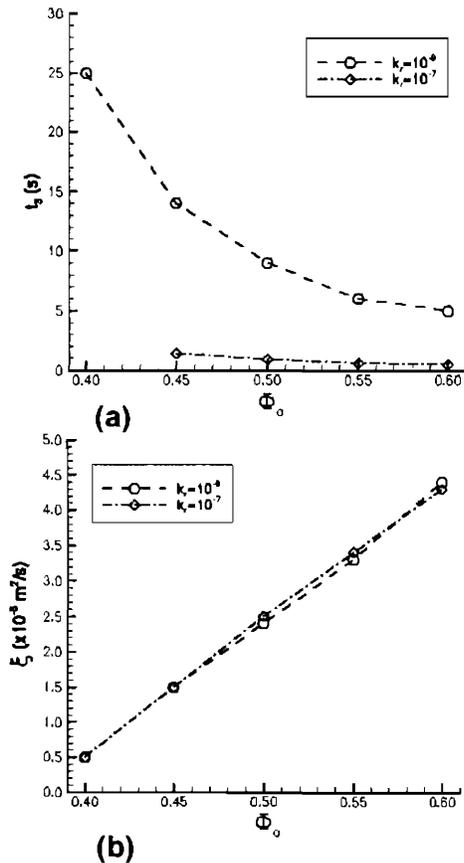


Figure 6. Influence of  $\Phi_0$  in the post-reaction-dominated regime: (a) time of entry into this regime; and (b) intrinsic infiltration rate.

The time at which the infiltration moved from the reaction-dominated regime to the third post-reaction-dominated regime,  $t_3$ , shown in Fig. 6(a), was estimated from infiltration profiles. The intrinsic infiltration rate,  $\xi$ , in this regime is presented in Fig. 6(b).

For  $k_r=0$ ,  $\xi=1.0 \times 10^{-4}$  m/s over the range of porosity. For  $k_r=10^{-8}$  m/s and  $10^{-7}$  m/s, we can see that  $\xi$  increases linearly with porosity. We might expect this, because the permeability is proportional to the porosity, while capillary pressure remains fixed. When chemical reaction occurs, the entry into the post-reaction-dominated regime is retarded as porosity decreases, reflecting the increased influence of the reaction-generated forces on the infiltration dynamics. However, once in the post-reaction-dominated regime, the permeability in the fully reacted zone behind the front is more influential on the infiltration dynamics than the reaction-generated forces.

### Effect of initial pore radius

Next, we will examine the influence of the initial pore radius on the infiltration process, holding the porosity fixed at  $\Phi_0=0.50$  and varying the reaction rate,  $k_r$ , to 0,  $10^{-9}$ ,  $10^{-8}$ ,  $10^{-7}$  m/s and the pore radius to 0.10, 0.14, 0.17 and 0.20  $\mu\text{m}$ . The capillary pressure corresponding to these pore radii ranges from  $5.94 \times 10^6$  Pa to  $1.88 \times 10^7$  Pa (see Eq. (11)). For clarity, the infiltration curves are presented in sets by reaction rate in Fig. 7 for the first 30 seconds of the infiltration.

Decreasing pore radius retards the infiltration, as observed by Einset (1996) experimentally. Ignoring the chemical reaction for the moment, this would be expected since the capillary pressure is inversely proportional to  $r_0$ , while the permeability is proportional to  $r_0^2$ . Unlike the parameterization in porosity, the curves for  $k_r=0$  and  $10^{-9}$  m/s do not collapse onto one another in Figs. 7(a) and (b). We also see the effects of chemistry in the curves for  $k_r=10^{-8}$  and  $10^{-7}$  m/s in Figs. 7(c) and (d).

The curves for  $k_r=10^{-9}$  m/s (Fig. 7(b)) are nearly equivalent to the corresponding nonreacting case for the first second or two. All of the preforms at this reaction rate are fully saturated by 30 seconds except for  $r_0=0.10$   $\mu\text{m}$ . The intrinsic infiltration rate,  $\xi$ , shown in Fig. 7(f) is not constant for any of the pore radii over this time interval.

The characteristic bump in the infiltration profile due to the presence of the reaction-dominated regime is seen most clearly in Fig. 7(c) for  $k_r=10^{-8}$  m/s under 10 seconds. The corresponding slope of  $y_f^2(t)$  in Fig. 7(g) is not linear during this time. The infiltration has moved beyond the reaction-dominated regime by 10 seconds in Fig. 7(c). This occurs more quickly for the higher reaction rate in less than a second in Fig. 7(d). From Figs. 7(g) and (h), we can see that  $\xi$  appears constant once the infiltration process has left the reaction-dominated regime.

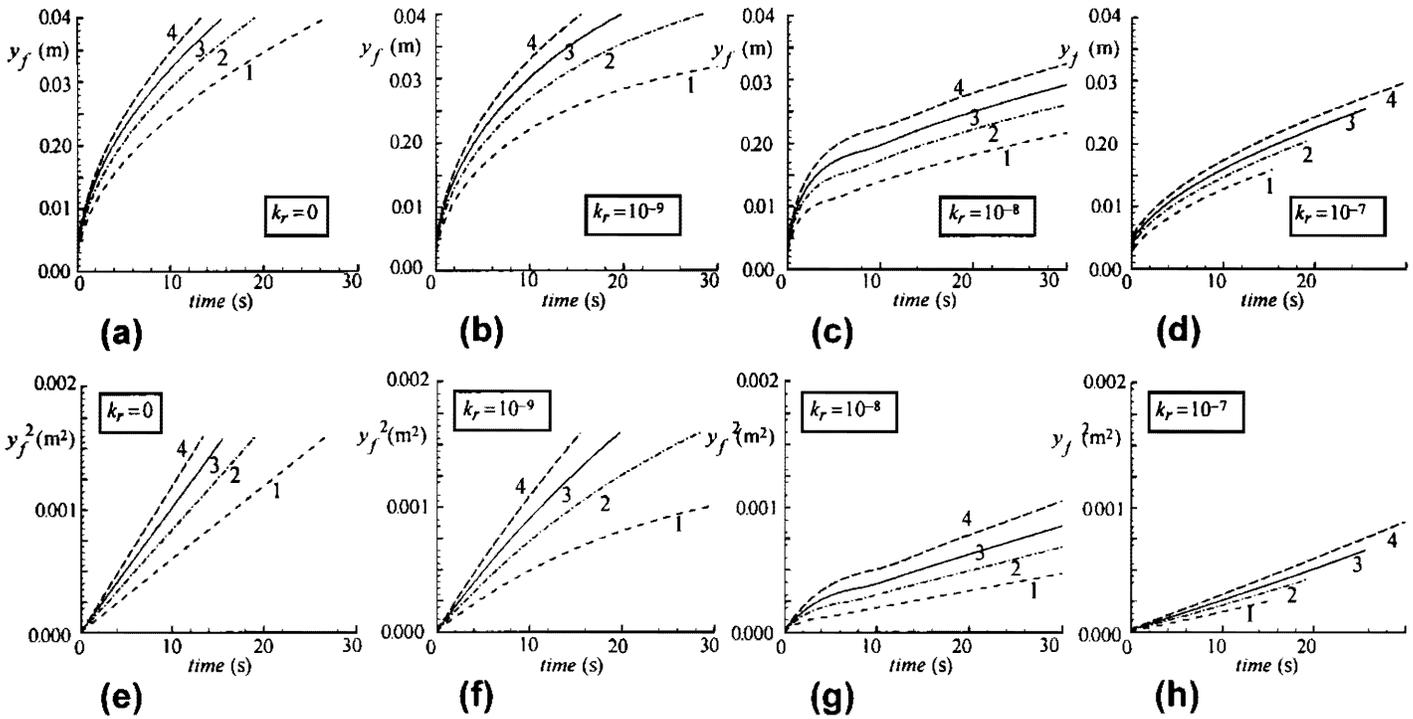


Figure 7. Effect of  $r_0$  on infiltration profiles for  $\Phi_0 = 0.50$ ;  $r_0 = 0.10$  (1), 0.14 (2), 0.17 (3) and 0.20 (4)  $\mu\text{m}$ : (a)  $y_f(t)$ ,  $k_r = 0.0$ ; (b)  $y_f(t)$ ,  $k_r = 10^{-9}$  m/s; (c)  $y_f(t)$ ,  $k_r = 10^{-8}$  m/s; and (d)  $y_f(t)$ ,  $k_r = 10^{-7}$  m/s; (e)  $y_f^2(t)$ ,  $k_r = 0.0$ ; (f)  $y_f^2(t)$ ,  $k_r = 10^{-9}$  m/s; (g)  $y_f^2(t)$ ,  $k_r = 10^{-8}$  m/s; and (h)  $y_f^2(t)$ ,  $k_r = 10^{-7}$  m/s.

The dependence of the total infiltration and conversion times on pore radius is shown in Fig. 8. A wide gap is once again apparent between  $t_{e,i}$  and  $t_{e,r}$  for  $k_r = 10^{-9}$  m/s (unfilled and filled circles). For the lowest pore radius and reaction rate (filled circle, left of arrow), the infiltration process had moved well beyond the reaction-dominated regime when the upper boundary was encountered. On the other hand, much of the conversion took place in a fully saturated preform for the larger pore radii (filled circles, right of arrow). Recall that, for a fully saturated preform, the pore radius decreases linearly with time, Eq. (1), to complete conversion. The trend of increasing  $t_{e,r}$  with increasing radius at high pore radii therefore indicates that, under these conditions, the absence of the front reduces the effectiveness of the conversion.

Figure 9 shows the influence of pore radius on the start of the post-reaction-dominated regime and on the intrinsic infiltration rate. Fig. 9(a) indicates that decreasing pore radius reduces the temporal extent of the reaction-dominated regime, an effect which is most pronounced at the lower reaction rate. Unlike the porosity-dependent parameterization,  $\xi$  is not constant for  $k_r = 0$  (squares), but rather increases with increasing pore radius, due to increasing permeability. As before, we should expect this because the capillary pressure is inversely proportional to  $r_0$ , while the permeability is

proportional to  $r_0^2$ . Fig. 9(b) indicates that  $\xi$  is linearly proportional to  $r_0$  and is not affected greatly by the reaction rate. These results imply that the infiltration in the post-reaction-dominated regime is governed primarily by the permeability of the preform in the reacted zone behind the front and by the capillary pressure.

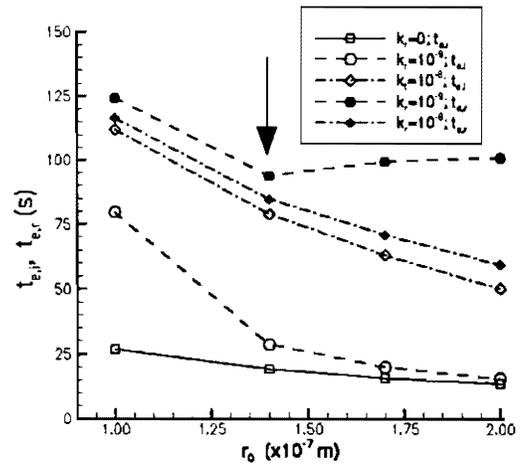


Figure 8. Influence of  $r_0$  on required time for complete infiltration (unfilled symbols) and complete conversion (filled symbols) for  $k_r = 0$ ,  $\square$ ;  $10^{-9}$ ,  $\circ$ ; and  $10^{-8}$  m/s,  $\blacklozenge$ .

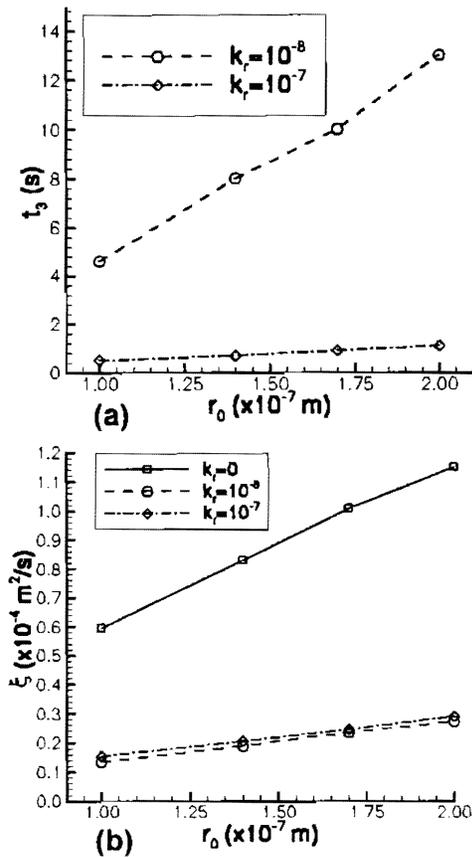


Figure 9. Influence of  $r_0$  in the post-reaction-dominated regime: (a) time of entry into this regime; and (b) intrinsic infiltration rate.

## CONCLUSIONS

We have examined the infiltration of liquid Si into a constant-volume microporous C preform to form a monolithic SiC ceramic composite for the range of  $\Phi_0 = 0.40-0.60$ ,  $r_0 = 0.10-0.20$   $\mu\text{m}$ , and  $k_r = 0, 10^{-9}, 10^{-8}, 10^{-7}$  m/s. The initial pore radius and porosity were uniform over the entire preform.

We have seen that the infiltration dynamics are dependent on reaction rate, initial porosity and initial pore radius. Significantly, it is also a function of the flow regime arising from the competition between pressure and reaction forces. We did not find a simple correlation for the time and distance associated with the reaction-dominated regime. However, numerical simulation can be used to predict these details.

Initially, the pressure forces dominated as the infiltrant impulsively entered the preform. Later, the presence of the chemical reaction retarded the infiltration due to a competition between the pressure forces and the volumetric forces due to the chemical conversion. When the reactants in the entry to the preform were consumed, the reaction zone was confined

to a small region behind the front. Furthermore, the infiltration rate increased, although it was significantly smaller than that of the corresponding nonreacting case.

In the first pressure-dominated regime and the early stages of the second reaction-dominated regime, a porosity increase of 50% had a negligible effect on the infiltration profiles, as seen in the data for  $k_r = 0$  and  $10^{-9}$  m/s.

The reaction-generated forces may impede or enhance the flow, depending on the density variation among the products and reactants. For the SiC/SiC system with a dense product, the conversion was shown to retard the flow. The temporal and physical extent of the reaction-dominated regime decreased with increasing reaction rate, decreasing pore radius and increasing porosity. In this regime, the velocity within the preform was retarded significantly due to the curvature of the pressure profiles, causing a characteristic bump in the infiltration profiles. Theoretically, the flow can become stationary at the front; this is to be distinguished from choking in that this is a temporary phenomenon. For  $k_r = 10^{-8}$  m/s, the reaction-dominated regime was in evidence from about 2 to 10 seconds, while it was completed by one second for  $k_r = 10^{-7}$  m/s. The intrinsic infiltration rate was not constant in this regime, indicating that the location of the front did not have a square-root dependence on time. Aside from the implications on infiltration dynamics, the maximum process temperature was found, in previous work, to occur in this regime. The reaction-dominated regime ended soon after the carbon in the initial entry zone was completely consumed.

Beyond the reaction-dominated regime, the infiltration settled into a simpler pattern. The reaction zone was localized to an area just behind the front. The intrinsic infiltration rate was constant and proportional to the initial porosity and to the initial pore radius. Although the permeability is proportional to the square of the pore radius, the capillary pressure is inversely proportional to the pore radius, accounting for this behavior. The intrinsic infiltration rate was not greatly influenced by the magnitude of the reaction rate in this regime. We therefore deduce that the infiltration was primarily affected by the capillary pressure at the front and the permeability in the fully reacted zone behind the front. Consequently, the intrinsic infiltration rate can be characterized by a constant beyond the reaction-dominated regime, although the line representing  $\xi t$  must be offset from the origin to coincide with the data.

For this system, numerical simulation of the parameter space provides the best prediction of optimal parameters for minimum conversion time. For an actual process, we should also include preform length as a parameter, as that will affect whether or not the process leaves the reaction-dominated regime. For  $k_r = 10^{-9}$  m/s, the infiltration was completed during the early stages of the reaction-dominated regime, except at the smallest pore radii. If the entire preform

reached saturation during this time, the overall conversion time was seen to increase with increasing pore radius, since the evolving pore radius decreased strictly linearly with time, as set by the chemistry of the system. In contrast, if the infiltration moved into the post-reaction-dominated regime, as seen at lower pore radius, the conversion time decreased with increasing pore radius. The presence of the advancing front, with its accompanying complex infiltration dynamics, can either improve or reduce the efficiency of the conversion process, depending on the initial parameters. If the infiltration was allowed to proceed through the sequence of infiltration regimes, the overall conversion time decreased with increasing porosity, pore radius and reaction rate.

This sort of analysis could provide insight into the design of functionally graded materials. For example, local modification of the pore radius or porosity in the post-reaction-dominated regime could be used to speed up the infiltration rate. Such a modification in the reaction-dominated regime could have major implications on the efficiency of the process. The advantage of a numerical simulation is that it could range over the desired parameter space with minimal effort, allowing for more precise optimization.

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